

## Towards an Er-doped Si nanocrystal sensitized waveguide laser – the thin line between gain and loss

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### Abstract

Important progress is being made in the development of a Si based waveguide laser operating at 1.5  $\mu\text{m}$ . The gain medium responsible for the recent progress is Er-doped Si nanocrystal co-doped  $\text{SiO}_2$ , a composite material that can potentially be fabricated using a VLSI compatible process. The material combines the efficient broad absorption spectrum of Si nanocrystals with the efficient narrow linewidth emission of Er ions. This combination promises to enable the fabrication of a broad band pumped integrated optical amplifier or laser in a Si based materials system. In this paper we systematically discuss the applicability of Si nanocrystals to serve as sensitizers for Er, relating the available data to key sensitizer requirements.

### 1. Introduction

Current Er-doped planar waveguide amplifiers and lasers make use of a pump laser emitting at 0.98  $\mu\text{m}$  that excites  $\text{Er}^{3+}$  ions into their second excited state ( $^4I_{11/2}$ ). From this level, the ions quickly relax non-radiatively to the long lived first excited state ( $^4I_{13/2}$ ) providing the population inversion necessary for gain (see Figure 1(a)). For an Er doped waveguide laser, this means an accurately tuned semiconductor pump laser needs to be integrated with an optical waveguide. This is an expensive approach, since it requires an

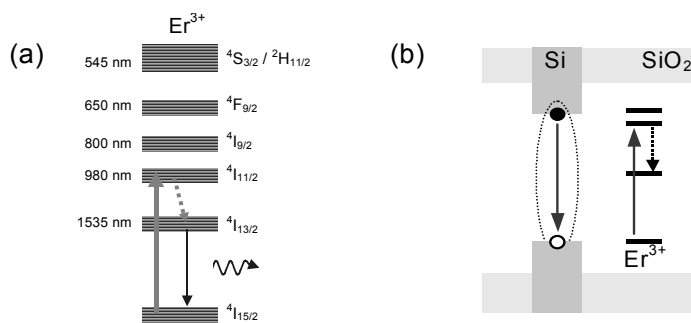


Figure 1(a) Schematic band structure of  $\text{Er}^{3+}$  showing excitation at 980 nm followed by non-radiative relaxation to the first excited state and emission at 1.5  $\mu\text{m}$  (b) Schematic band structure of an Er doped Si nanocrystal in  $\text{SiO}_2$ , showing the recombination of a quantum confined electron-hole pair and simultaneous Er excitation

expensive pump laser and a difficult alignment procedure to obtain optimum coupling between pump laser and waveguide. If the pump laser could somehow be eliminated from this scheme, the fabrication of low-cost Si based Er doped optical amplifiers and lasers operating at 1.5  $\mu\text{m}$  would become a real possibility. This is exactly what co-doping with Si nanocrystals promises to do.

It has been observed by several groups<sup>1-6</sup> that Er can be excited indirectly in Si nanocrystal doped  $\text{SiO}_2$ . The excitation occurs via optically generated electron-hole pairs inside Si nanocrystals that recombine and transfer energy to the Er ions. Figure 1(b) shows a schematic band diagram of  $\text{SiO}_2$  containing a Si nanocrystal and  $\text{Er}^{3+}$ . First a photon is absorbed by the nanocrystal, which causes generation of an exciton inside the nanocrystal. This exciton can recombine radiatively, emitting a photon with an energy that depends on the nanocrystal size. If an Er ion is present close to the nanocrystal, the exciton can recombine non-radiatively by bringing Er into one of its excited states. Si nanocrystals can be excited in a broad wavelength range ( $\lambda < \sim 1 \mu\text{m}$ ) and have a pump absorption cross-section that is several orders of magnitude higher than the cross-section for direct Er excitation. These two features make codoping with Si nanocrystals a promising route to fabricating integrated waveguide amplifiers and lasers that can be pumped with low-power broadband light sources.

Er excitation via Si nanocrystals must meet several additional requirements to be of use in an optical amplifier. In this article we show measurements that demonstrate (a) the large nanocrystal absorption cross-section  $\sigma_{\text{abs}} > 10^{-16} \text{ cm}^2$ , (b) a high energy transfer efficiency  $\eta \geq 55\%$ , (c) the broad absorption band of Si nanocrystals, (d) a high transfer rate  $R \geq 10^6 \text{ s}^{-1}$ , (e) a limit to the excitable Er concentration of  $\sim 1-2$  Er ions per nanocrystal, (f) evidence for free-carrier induced absorption in Si nanocrystal doped optical waveguides, and (g) an unexpected enhancement of the direct optical absorption cross-section of Er in a nanocrystal doped medium. We discuss how all these observations affect the potential use of Er doped Si nanocrystal co-doped  $\text{SiO}_2$  as an optical gain medium. Finally, calculations will be presented that offer an indication of the net gain that may be expected in this type of waveguides amplifiers.

## 2. Er sensitizers – requirements

A potential sensitizer for Er doped waveguide lasers should satisfy a number of requirements. In this Section we list all necessary requirements, and in Section 4 we will discuss how Si nanocrystals perform in these areas.

### 2.1 Absorption cross-section

The sensitizer should have a large absorption cross-section to allow for Er population inversion at relatively low pump intensities. It should be noted here that a large cross-section is not always desirable, since a too large cross-section can in some cases cause all pump power to be absorbed in a short part of the waveguide, reducing the useful length of the waveguide. For extremely large cross-sections however, a significant fraction of the pump power can be absorbed when illuminating the waveguide at normal incidence,

removing the need for accurate pump coupling. This is an important technological advantage that, as we will show in Section 4.1, can be achieved with Si nanocrystals.

## *2.2 Excitation efficiency*

Absorbed energy should be transferred to Er with high quantum efficiency and high power efficiency. In the case of nanocrystals we will show that the quantum efficiency is larger than 55%. However, since the power efficiency depends on the excitation wavelength. When a wavelength of 458 nm is used, these 2.7 eV photons result in a 0.8 eV Er excited state, with the excess 1.9 eV transformed into heat. This means that at a quantum efficiency of 55% the power efficiency is only 16%. Consequently, in some cases it may be advantageous to excite at longer wavelength. Note however that at longer wavelength the Si absorption cross-section is reduced.

## *2.3 Absorption linewidth*

The pump source is one of the main contributions to the cost of Er doped amplifiers, since it has to perfectly match the narrow Er absorption band at 980 nm. If a sensitizer is used, the pump wavelength needs only to be matched to the absorption of the sensitizer. In case of a sufficiently wide absorption band, more cost efficient broad-band pump sources such as LEDs can be used for excitation, allowing for a significant cost reduction of the laser or amplifier.

## *2.4 Transfer rate*

In any laser or amplifier, the total possible output power is determined by the rate at which the amplifying transition is pumped. In the case of a sensitized Er transition the energy transfer rate from the sensitizer to Er could form an excitation bottleneck. Once the transfer rate has been determined, an estimate of the maximum output power can be made, as will be shown in Section 4.4.

## *2.5 Excitable fraction*

When using an Er sensitizer, it is essential that all Er ions can be excited. This is especially important for three level systems such as the 1.5  $\mu\text{m}$  transition of Er. In  $\text{Er}^{3+}$  the amplifying transition occurs from the first excited state to the ground state, and consequently any fraction of non-excitable Er will efficiently absorb at the signal wavelength. Note that if the amplifying center is a four-level system in which the relevant transition occurs between *two* excited states, the problem of signal absorption is completely removed.

## 2.6 Induced loss

Any waveguide loss introduced by the sensitizer will reduce the maximum output power of the sensitized laser or amplifier. Possible loss contributions include increased scattering due to an inhomogeneous sensitizer distribution, changes in the materials processing that introduce waveguide roughness, and introduction of free charges that can lead to free carrier absorption. The latter effect is of special relevance to the case of Si nanocrystal sensitization, since the nanocrystal excitation requires the generation of electron-hole pairs, which can in fact absorb at 1.5  $\mu\text{m}$  wavelength. In bulk Si, free carrier absorption cross-sections are as large as  $10^{-17} \text{ cm}^2$  around 1.5  $\mu\text{m}$ .<sup>7</sup> This is more than a factor 1000 higher than a typical Er emission cross-section, meaning that the presence of one exciton for every 1000 Er ions could completely suppress any amplification. It should be noted that free carrier absorption cross-sections in Si nanocrystal doped  $\text{SiO}_2$  may be reduced compared to those in bulk Si due to quantum size effects.

## 3. Experimental

This paper contains results on two different sample types. The first part deals with thin oxide layers doped with Er and Si nanocrystals by means of ion implantation. For these samples, a 100 nm thick layer of  $\text{SiO}_2$  was grown on a lightly B-doped Si(100) substrate by means of wet thermal oxidation. This layer was implanted with 35 keV Si to a dose of  $6 \times 10^{16} \text{ cm}^{-2}$ . The implantation yields an approximately Gaussian depth distribution of excess Si in the  $\text{SiO}_2$  film, with a peak concentration of 19 at.% (corresponding to a composition of  $\text{Si}_{0.46}\text{O}_{0.54}$ ) at a depth of 45 nm. The samples were subsequently annealed at 1100 °C for 10 min. in vacuum at a base pressure below  $3 \times 10^{-7}$  mbar in order to induce nucleation and growth of Si nanocrystals. This treatment has been shown to produce Si nanocrystals with a diameter in the range 2-5 nm.<sup>8</sup> Assuming a typical nanocrystal diameter of 3 nm ( $\sim 10^3$  Si atoms), the nanocrystal peak concentration is estimated to be  $\sim 10^{19} \text{ cm}^{-3}$  at the center of the  $\text{SiO}_2$  layer. The samples were then implanted with different Er doses in the range  $3.6 \times 10^{13} \text{ cm}^{-2} - 5.1 \times 10^{15} \text{ cm}^{-2}$  at a fixed energy of 125 keV. These implants result in an approximately Gaussian Er depth distribution, with Er peak concentrations ranging from 0.015 at.% to 1.8 at.% at a depth of 61 nm. A non Er-implanted sample was kept as a reference. After the Er implants, all samples (including the non-Er implanted) were annealed for 10 min. in vacuum at 1000 °C to remove implantation-induced damage. In order to further reduce defect-related luminescence and to saturate dangling bonds on the Si nanocrystal surface, a passivating anneal was performed at 780 °C for 30 min. in forming gas ( $\text{H}_2:\text{N}_2$  at 1:9) at atmospheric pressure.

The second series of experiments involve the fabrication of Er doped nanocrystal sensitized waveguide samples in  $\text{SiO}_2$ . Silicon ions were implanted at 165 keV to a fluence of  $1.7 \times 10^{17} \text{ cm}^{-2}$  into a 5  $\mu\text{m}$  thick layer of  $\text{SiO}_2$  that was grown by wet thermal oxidation of Si(100). The material was annealed at 1100 °C for 10 min. in flowing Ar in order to induce nucleation and growth of Si nanocrystals. The temperature of 1100 °C was found to yield maximum nanocrystal luminescence. This material was implanted with Er at 700 keV to a fluence of  $1.2 \times 10^{16} \text{ cm}^{-2}$  and annealed at 1000 °C for 10 min. in flowing Ar to remove implantation induced damage. A sample containing no Er was kept as a reference. These samples were annealed at 800 °C for 10 min. in forming gas ( $\text{H}_2:\text{N}_2$  at 1:9) to optimize the

nanocrystal and Er photoluminescence intensity. The excess Si concentration profile peaks at a depth of 240 nm, and has a full width at half maximum (FWHM) of 160 nm. The Si peak concentration is 42 at.%, corresponding to an excess Si concentration of 13 at.%. Assuming an average nanocrystal diameter of 3 nm, the nanocrystal concentration at the peak of the Si profile is estimated to be  $1.3 \times 10^{19} \text{ cm}^{-3}$ . The Er concentration profile peaks at a depth of 240 nm, with an Er peak concentration of 1.3 at.% and a FWHM of 130 nm.

A series of 3.5  $\mu\text{m}$  wide ridge waveguides was formed in the implanted and annealed  $\text{SiO}_2$  film using standard photolithography and Ar beam etching to a depth of 0.5  $\mu\text{m}$ , i.e. well beyond the Si doped region. The excess Si inside the ridges locally raises the index of refraction, providing the index contrast required for transverse and lateral optical mode confinement. To reduce scattering losses, the waveguides were covered with a 1.25  $\mu\text{m}$  thick  $\text{SiO}_2$  cladding layer using microwave sputtering. The samples were subsequently cut to a length of 8 mm, and the waveguide input and output facets were mechanically polished.

Photoluminescence (PL) measurements on planar samples were taken using the 458 nm line of an Ar laser as excitation source at a peak power of 1 mW in a  $\sim 1 \text{ mm}^2$  laser spot. The laser beam was modulated on-off at 11 Hz using an acousto-optical modulator. The emitted light was passed through a grating monochromator and detected using standard lock-in techniques. Spectra were measured in the range 600-1150 nm using a AgOCs photomultiplier tube (PMT), and in the range 1100-1700 nm using a liquid-nitrogen cooled Ge detector. All spectra were corrected for the detector response. Photoluminescence decay traces were recorded using a multichannel photon counting system in combination with the PMT and a digitizing oscilloscope in combination with the Ge detector. The system response for the two cases was 150 ns and 160  $\mu\text{s}$ , respectively. Except when noted otherwise, measurements on the planar samples were performed at 15 K using a closed-cycle He cryostat.

Optical transmission measurements on the waveguide samples were performed using a fiber-coupled white light source emitting in the 1000-1800 nm range, which was modulated on-off at 270 Hz for lock-in detection. The white light was butt-coupled to the nanocrystal waveguides using a single mode tapered fiber. The transmitted light was collected by imaging the waveguide output facet onto a multimode fiber using a 10 $\times$  microscope objective. The collected light was led to a grating monochromator and detected using a liquid nitrogen cooled Ge detector, in combination with standard lock-in techniques. Optical mode images were obtained by coupling 1.49  $\mu\text{m}$  light from an InGaAsP diode laser into the waveguide, and imaging the waveguide output facet on an infrared camera using a 10 $\times$  microscope objective.

For gain measurements the Si nanocrystal and Er doped waveguides were optically pumped CW using the 458 nm line of an Ar laser. The pump light was projected onto the waveguides from the top using a cylindrical lens. The elongated spot was aligned with the waveguide, and covered the full waveguide length. In this way the pump power was evenly distributed over the full length of the waveguide. In these measurements the modulated white light source was coupled into the waveguide, and the transmitted signal around 1.5  $\mu\text{m}$  was detected as a function of pump power using lock-in detection.

## 4. Results and discussion

Figure 2 shows a PL spectrum of a SiO<sub>2</sub> film doped with Si nanocrystals (marked "no Er"). The sample shows a broad luminescence band peaking at 790 nm. This luminescence is caused by the radiative recombination of electron-hole pairs (excitons) confined within the Si nanocrystals. Due to quantum confinement<sup>9</sup> the exciton luminescence appears at energies above the bandgap energy of bulk Si (1.17 eV at 15 K). The large spectral width of the nanocrystal luminescence in these samples is caused mainly by the broad nanocrystal size distribution (2-5 nm diameter). The luminescence peak at 1.13 μm is caused by phonon-assisted electron-hole pair recombination in the Si substrate. Figure 2 also shows luminescence spectra for samples containing Er at various concentrations. The incorporation of 0.015 at.% Er reduces the nanocrystal luminescence by more than a factor two, and a luminescence peak appears at a wavelength of 1.536 μm. This wavelength corresponds to the radiative transition from the first excited state (<sup>4</sup>I<sub>13/2</sub>) to the ground state (<sup>4</sup>I<sub>15/2</sub>) of Er<sup>3+</sup>. Increasing the Er concentration leads to a further reduction of the nanocrystal luminescence intensity, accompanied by an increase of the Er luminescence intensity. The inverse correlation between nanocrystal emission and Er emission is a first indication that Si nanocrystals act as sensitizers for Er. Since the applied excitation wavelength does not overlap with one of the Er absorption lines, the appearance of Er luminescence provides further evidence for sensitized luminescence. In the following we will look in detail at the behavior of these two luminescence features, and the results will be used to evaluate the potential of Si nanocrystals as Er sensitizers.

### 4.1 Absorption cross-section

In order to determine the nanocrystal absorption cross-section, we measured the nanocrystal luminescence  $1/e$  rise time ( $\tau_r$ ) and decay time ( $\tau$ ) at 5 mW pump power (see time dependent data in Fig. 3(a)). From these two time constants we can deduce the excitation

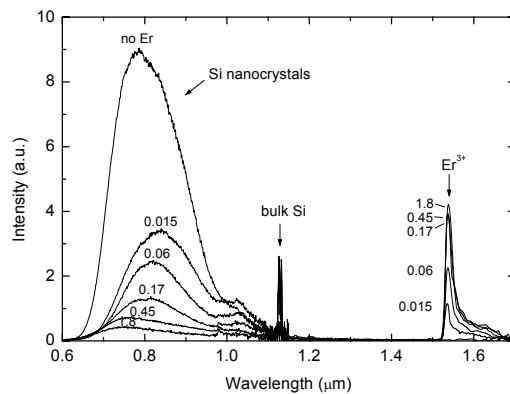


Figure 2. Photoluminescence spectra of Si nanocrystal doped SiO<sub>2</sub> containing different Er concentrations in the range 0 - 1.8 at.%, measured at 15 K using a pump power of 1 mW at 458 nm.

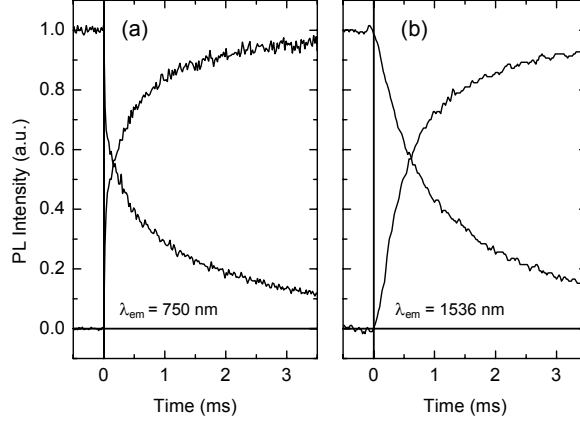


Figure 3. Luminescence rise time and decay measurements at 5 mW pump power at 15 K for (a) nanocrystal luminescence at 750 nm, and (b) Er luminescence at 1536 nm

rate  $R = (1/\tau_r - 1/\tau)$ , assuming that the nanocrystals effectively behave as a two-level system. At a pump intensity of  $\sim 5 \text{ mW/mm}^2$  we find for the nanocrystal excitation rate  $R_{nc} = \sigma_{nc} \times \Phi_{\text{phot}} = 1040 \text{ s}^{-1}$  with  $\sigma_{nc}$  the nanocrystal absorption cross-section and  $\Phi_{\text{phot}}$  the photon flux. Inserting the known photon flux of  $\sim 1 \times 10^{18} \text{ cm}^{-2}\text{s}$  we find  $\sigma_{nc} = 9 \times 10^{-16} \text{ cm}^2$ . This value is several orders of magnitude larger than direct Er optical absorption cross-sections, which are typically in the  $10^{-21} \text{ cm}^2$  range. Note that the value obtained is somewhat larger than other values reported at this excitation wavelength, see e.g. Künzer *et al.*<sup>10</sup> ( $\sigma_{\text{abs},458\text{nm}} \approx 4 \times 10^{-16} \text{ cm}^2$ ) and Priolo *et al.*<sup>11</sup> ( $\sigma_{\text{abs},488\text{nm}} \approx 1 \times 10^{-16} \text{ cm}^2$ ). This kind of discrepancy may be inherent to the type of measurement: the cross-section depends on the nanocrystal diameter to the  $\sim 3^{\text{rd}}$  power, and the obtained value for the effective cross-section depends quadratically on the estimated laser beam diameter.

As was mentioned in the introduction, the large absorption cross-section of Si nanocrystals may allow for unconventional pumping geometries, such as normal-incidence pumping, or side-illumination in which the whole waveguide length is illuminated by a source that is coupled into a slab-waveguide mode. For this type of geometry to be effective, a significant fraction of the pump power should be absorbed within a typical waveguide thickness. To see whether this is possible we can estimate the absorption depth in typical Si nanocrystal doped samples. The absorption depth can be obtained from the absorption coefficient  $\alpha(\text{cm}^{-1}) = \sigma_{\text{abs}} \times C_{\text{nc}}$ . If we take a conservative absorption cross-section of  $10^{-16} \text{ cm}^2$  and a typical nanocrystal concentration of  $10^{19} \text{ cm}^{-3}$ , we obtain an absorption depth  $d_{1/e} = 1/\alpha = 10 \text{ }\mu\text{m}$ . Consequently, pumping at a wavelength of 458 nm, 63% of the pump power can be absorbed in a 10  $\mu\text{m}$  thick waveguide. Pumping at shorter wavelengths will further reduce the required waveguide thickness. This shows that pumping either laterally or vertically is a realistic option in this type of material.

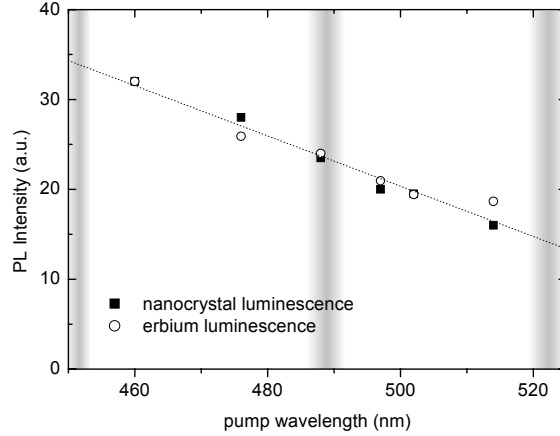


Figure 4. Excitation spectrum of the nanocrystal luminescence at 750 nm and the Er luminescence at 1536 nm. The dotted line is a guide to the eye. The vertical bands indicate the position of direct Er optical absorption lines.

#### 4.2 Excitation efficiency

In order to determine the Er excitation efficiency, we have also measured the Er luminescence  $1/e$  rise time ( $\tau_r$ ) and decay time ( $\tau$ ) on the same sample as discussed in Section 4.1 (see Fig. 3(b)). Under the same pump conditions we find for the Er excitation rate  $R_{Er} = 570 \text{ s}^{-1}$ , which is of the same order of magnitude as the previously obtained nanocrystal excitation rate  $R_{nc} = 1040 \text{ s}^{-1}$ . These numbers give an estimate of the energy transfer efficiency, or the internal quantum efficiency. Assuming each nanocrystal couples to only one Er ion, we conclude that  $570/1040 = 55\%$  of the generated excitons recombine by transferring energy to the Er ion.<sup>12</sup> The obtained 55% is a lower estimate for the excitation efficiency. For example, if we assume each nanocrystal couples to two Er ions, the transfer efficiency would be  $2 \times 570/1040 \approx 100\%$ . The requirement for efficient energy transfer is clearly satisfied. Note that it is difficult to argue that in this sample more than 2 Er ions couple to one nanocrystal, since this would lead to internal quantum efficiencies in excess of 100%. This apparent concentration limit will be further discussed in Section 4.5.

#### 4.3 Absorption linewidth

Figure 4 shows photoluminescence excitation spectra of the nanocrystal luminescence at 750 nm and the Er luminescence at 1536 nm. At fixed pump power, the nanocrystal luminescence intensity and the Er luminescence intensity are seen to increase by a factor two when the excitation wavelength is decreased from 514 nm to 458 nm. The absence of absorption resonances in the Er excitation spectrum indicates that the Er is not excited directly by absorption of pump photons, but indirectly via an energy transfer process from Si nanocrystals, as was already suggested above. The observed wavelength dependence of the luminescence intensity is attributed to the increasing nanocrystal optical absorption cross-section for increasing photon energy, leading to a higher exciton generation rate.



These results give direct evidence that Er in this material can be excited in a broad wavelength range, allowing for broad band pumping of amplifiers and lasers based on this materials system.

#### 4.4 Transfer rate

Figure 5 shows nanocrystal luminescence decay traces taken at 750 nm and  $T=15$  K for samples containing different Er concentrations. In a sample containing no Er, the nanocrystal luminescence shows a  $1/e$  lifetime of 2.0 ms. The decay is described by stretched exponential decay of the form  $I(t)=\exp(-(t/\tau)^\beta)$  with  $\tau=2.0$  ms and  $\beta=0.65$  (solid line). The same curve has been overlaid on all data in order to facilitate comparison between the different traces. Incorporation of Er leads to a significant reduction of the nanocrystal luminescence intensity at 750 nm, as was already observed in Fig. 2. The decay time however varies only slightly from sample to sample, and no trend is observed with increasing Er concentration. The sample containing the highest Er concentration shows a small initial fast decay, which is attributed to luminescence from defects in the  $\text{SiO}_2$  matrix. These lifetime measurements confirm the strong-coupling model,<sup>5</sup> which states that the Er induced nanocrystal PL intensity decrease in Fig. 2 is effectively due to a reduction in the number of luminescent nanocrystals. All remaining nanocrystal luminescence originates from nanocrystals that are not coupled to Er, which show their intrinsic decay characteristics. This model also explains the initially surprising observation that the Er luminescence intensity is virtually temperature independent, whereas the nanocrystal luminescence intensity varies by as much as a factor 4 in the temperature range 20 K - 300 K: in nanocrystals that do not couple to Er, temperature dependent radiative and non-radiative exciton recombination rates affect the luminescence efficiency, while in nanocrystals that couple to Er, the non-radiative transfer to Er is the dominant recombination process, occurring at a fixed efficiency over the full temperature range.

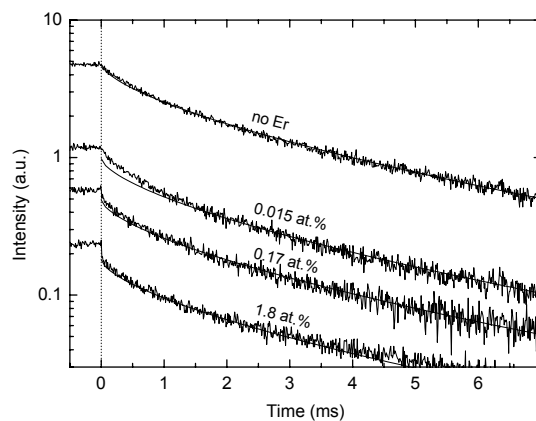


Figure 5. Photoluminescence decay traces of the nanocrystal luminescence at 750 nm for different Er concentrations in the range 0 - 1.8 at.%, measured at 15 K using 1 mW pump power at 458 nm. The pump is switched off at  $t=0$ . The drawn lines represent stretched exponential decay with  $\tau=2.0$  ms and  $\beta=0.65$ .

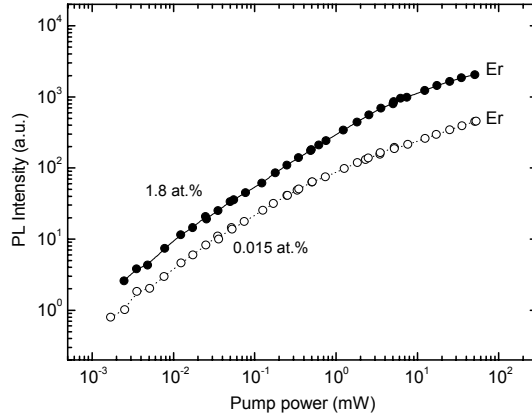


Figure 6. Photoluminescence intensity of the Er luminescence intensity as a function of pump power for samples containing 0.015 at.% and 1.8 at.% Er, measured at 15 K using 458 nm pump light in a  $\sim 1 \text{ mm}^2$  spot.

These observations also allow us to obtain an upper limit to the energy transfer rate constant, since for the energy transfer to be the dominant exciton recombination mechanism, it has to occur at a significantly higher rate than the spontaneous exciton recombination. The fastest observed nanocrystal decay time in these samples is 21  $\mu\text{s}$ , and consequently the energy transfer has to occur faster than  $\sim 1 \mu\text{s}$ .

Given the obtained value for the energy transfer rate of  $(1 \mu\text{s})^{-1} = 10^6 \text{ s}^{-1}$ , we can determine the maximum output power of an Er doped Si nanocrystal sensitized waveguide amplifier. Assuming a nanocrystal concentration of  $10^{19} \text{ cm}^{-3}$  and 2 excitable Er ions per nanocrystal in a  $10 \mu\text{m} \times 10 \mu\text{m} = 10^{-6} \text{ cm}^2$  waveguide, we find a maximum output power of  $2 \times 10^{19} \text{ cm}^{-3} \times 10^{-6} \text{ cm}^2 \times 10^6 \text{ s}^{-1} \times 0.8 \text{ eV} = 2.6 \text{ W/cm}$ . This shows that the transfer rate is sufficiently fast to allow for the amplification of high power signals over short waveguide lengths.

#### 4.5 Excitable fraction

Figure 6 shows the effect of pump power on the Er and nanocrystal luminescence intensity for samples containing 0.015 at.% Er and 1.8 at.% Er respectively. At pump powers below 20  $\mu\text{W}$  the Er luminescence in both samples depends linearly on pump power, and the Er luminescence intensities from the two samples differ by a factor  $\sim 2$ . Increasing the pump power produces a sub-linear increase of the Er luminescence in both samples, suggesting that a significant fraction of the excitable Er is brought into the first excited state, or that non-radiative processes come into play. At a pump power of 50 mW the Er luminescence intensity appears to level off. In this pump power regime the exciton generation rate is no longer the limiting factor for the Er luminescence intensity. Nevertheless, the Er luminescence intensities for the two samples at 50 mW pump power differ by only a factor 5, even though the total amount of Er in the samples differs by more than a factor 100.

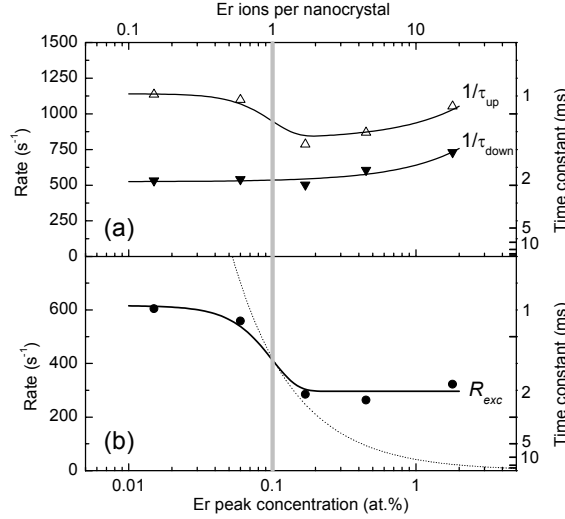


Figure 7.(a) Rise times ( $f$ ) and decay times ( $l$ ) of the  $1.536 \mu\text{m}$  Er luminescence, measured at 15 K using a pump power of 1 mW at 458 nm, and (b) the Er excitation rate derived from these data ( $l$ ). The drawn lines serve as a guide to the eye. The dotted line indicates a  $1/C_{Er}$  dependence.

The above shows that the concentration of excitable Er in the high concentration (1.8 at.%) sample is at most 5 times higher than in the low concentration (0.015 at.%) sample. This means that the concentration of excitable Er in these samples is  $< 0.1$  at.%. The existence of such a concentration limit is also observed in the Er excitation rate, as will be shown below.

When the number of Er ions coupled to a nanocrystal is increased, the Er excitation rate per ion can be expected to decrease, since several Er ions now compete for the same excitons. In order to determine the concentration dependent Er excitation rate, we performed rise time and decay time measurements of the Er luminescence at  $1.536 \mu\text{m}$ . At the applied pump power of 1 mW all samples show approximately exponential time dependencies. Figure 7(a) shows the measured rates  $W_{rise}=1/\tau_{rise}$  ( $f$ ) and  $W_{decay}=1/\tau_{decay}$  ( $l$ ) obtained by exponential fitting of the data.

The Er decay rate increases from  $500 \text{ s}^{-1}$  to  $700 \text{ s}^{-1}$  as the Er peak concentration is increased from 0.015 at.% to 1.8 at.%. This increase is attributed to a concentration quenching effect which is known to occur when rare earth ions are spaced closely enough to allow for energy exchange between neighboring ions. As a result, excitation energy can migrate<sup>13</sup> to neighboring ions, which may in turn be coupled non-radiatively to quenching sites, *e.g.* defects or OH groups present in the matrix. In a simple concentration quenching model, the Er decay rate increases linearly with Er concentration, which is indeed observed in Fig. 7(a). By measuring the slope of the decay rate data in Fig. 7(a), we estimate<sup>14</sup> that the concentration of quenching sites in the Er implanted Si nanocrystal doped  $\text{SiO}_2$  film is as low as  $10^{18} \text{ cm}^{-3}$ .

Data for the Er excitation rate  $R_{exc}$  calculated from  $\tau_{rise}$  and  $\tau_{decay}$  are shown in Fig. 7(b) ( $l$ ). At an Er concentration of 0.015 at.%, the Er excitation rate is  $600 \text{ s}^{-1}$ . Increasing the Er concentration to 0.17 at.% reduces the excitation rate by a factor 2. A further

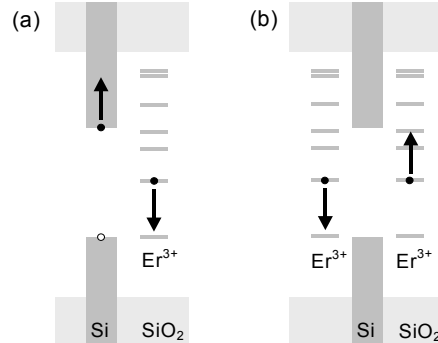


Figure 8. Schematic band diagram of Er and Si nanocrystal doped  $\text{SiO}_2$  showing the process of (a) Auger de-excitation, and (b) pair-induced quenching. These processes can account for the observed Er concentration limit in Si nanocrystal doped  $\text{SiO}_2$ . Note that the Er energy levels represent collective states of the  $\text{Er}^{3+}$  4f electrons, which are in fact well below the  $\text{SiO}_2$  valence band edge.

increase of the Er concentration has no effect on the excitation rate, even though at these concentrations several Er ions might couple to the same nanocrystal, which would reduce the excitation rate per Er ion. The fact that such a reduction is not observed, shows that there is an upper limit to the number of Er ions that can be excited by a single nanocrystal. From the data in Fig. 7(b) it is clear that this limit is reached at an Er concentration  $< 0.17$  at.% Er, which is consistent with the maximum value of 0.1 at.% Er found in the preceding paragraph.

The existence of an upper limit to the amount of excitable Er could indicate that there is a limit to the amount of *optically active* Er that can be incorporated in this material. This may for example be caused by Er clustering or  $\text{ErSi}_2$  formation, which would prevent the Er from being in the 3+ valence state. However, measurements on Er and Si nanocrystal doped waveguides show a strong  $\text{Er}^{3+}$  related absorption,<sup>15</sup> suggesting that a large fraction of the Er is in the optically active state. Alternatively, the observed concentration limit could be an intrinsic property of the excitation process. The amount of excitable Er will be low when the effective Er excitation efficiency is influenced by the presence of *excited* Er. Such a situation arises when the formation of an exciton near an excited Er ion leads to

- a) *Auger de-excitation*, in which an excited Er ion transfers energy to an exciton present in the nanocrystal (Fig. 8(a)). This process has been shown to occur in Er doped bulk Si.<sup>16</sup> After such Auger de-excitation the exciton can relax and subsequently excite an Er ion, effectively bringing the system back to the situation before the exciton was formed.
- b) *pair-induced quenching*. At sufficiently high Er concentration, two excited Er ions can interact yielding one Er ion in the  $^4I_{9/2}$  state, which rapidly decays to the first excited state, and one Er ion in the ground state (Fig. 8(b)). This co-operative upconversion effect usually produces a shortening of the Er decay rate at high

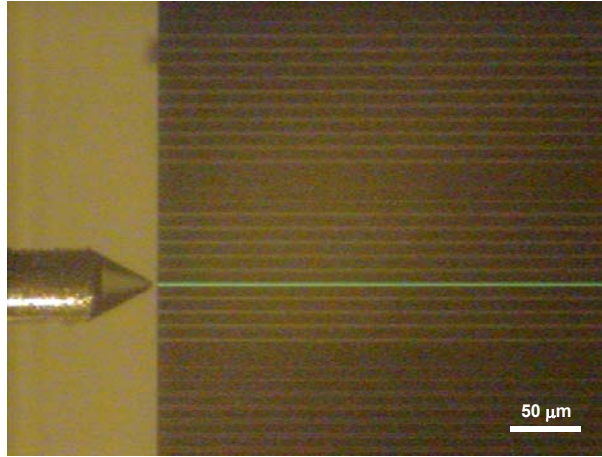


Figure 9. (color) Top view of an array of Er doped Si nanocrystal based channel waveguides, one of which is pumped with  $\sim 10$  mW at a wavelength of  $1.49 \mu\text{m}$  using a tapered optical fiber. The bright (green) emission from the waveguide is caused by radiative relaxation from the  $^4S_{3/2}$  and  $^2H_{1/2}$  levels that are populated in a two-step co-operative upconversion process.

pump powers, which has not been observed. However, if the Er-Er coupling is sufficiently strong, no effect on the lifetime is seen. This special case is called pair-induced quenching.

Evidence for upconversion has been found in Er doped Si nanocrystal sensitized optical waveguides when pumped at  $1.49 \mu\text{m}$  (see Fig. 9), and in planar samples containing Er and Si nanocrystals pumped at visible wavelengths.<sup>17</sup> The latter observation suggests that a high concentration of excited Er can be obtained in this type of material.

#### 4.6 Induced losses

Figure 11 shows iso-intensity contours of two optical mode images taken using  $1.49 \mu\text{m}$  light from an InGaAsP laser guided through the waveguide. The emission from the output facet was projected onto an infrared camera using a microscope objective. At this wavelength both waveguides only support the fundamental mode. For the smallest waveguide (width  $1 \mu\text{m}$ ) the mode is observed to be approximately circular, with a full width at half-maximum (FWHM) of  $2.8 \mu\text{m}$  in the  $x$ -direction, and  $3.0 \mu\text{m}$  in the  $y$ -direction. For the  $3.5 \mu\text{m}$  wide waveguide, the mode is observed to be slightly elliptical, with a full width at half-maximum (FWHM) of  $4.5 \mu\text{m}$  in the  $x$ -direction, and  $3.5 \mu\text{m}$  in the  $y$ -direction. From these modes and the known Er depth profile, an effective mode overlap  $\Gamma$  with the Er doped region can be obtained. For the  $3.5 \mu\text{m}$  waveguide this overlap is found to be 1%.

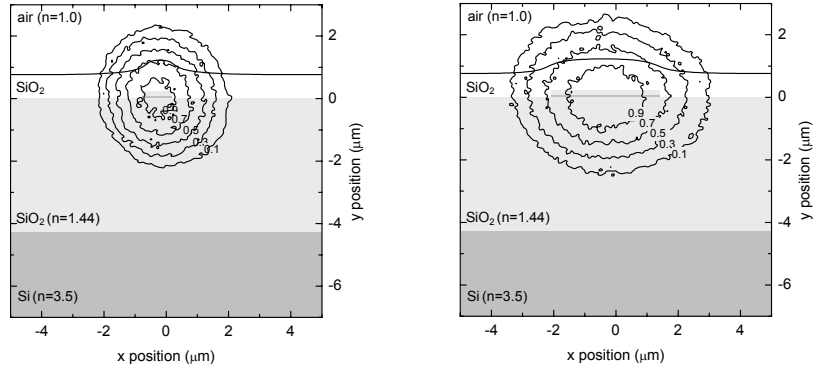


Figure 11. Optical mode images at a wavelength of 1.49  $\mu\text{m}$  taken at the output facet of two Er doped Si nanocrystal based ridge waveguides with widths of 1  $\mu\text{m}$  (a) and 3.5  $\mu\text{m}$  (b) respectively. The contour lines indicate constant intensities (0.2, 0.4, 0.6, and 0.8). A sketch of the waveguide structure is included

Figure 10 shows normalized transmission spectra of Si nanocrystal waveguides containing no Er (dashed line) and 1.3 at.% Er (solid line), measured using a fiber coupled white light source butt-coupled to the nanocrystal waveguide using a single-mode tapered fiber. In the Er-doped sample a clear dip is observed around 1.53  $\mu\text{m}$ , due to the  $^4I_{15/2} \rightarrow ^4I_{13/2}$  absorption transition of  $\text{Er}^{3+}$ . The peak absorption at 1.532  $\mu\text{m}$  is found to be 2.7 dB/cm. From this value and the measured values for  $\Gamma$  and  $N_{\text{Er}}$  we obtain an absorption cross-section  $\sigma_{\text{Er}}(1.532 \mu\text{m}) = 8 \times 10^{-20} \text{ cm}^2$ . This value is more than a factor ten larger than values found in Er doped Si ( $2 \times 10^{-20} \text{ cm}^2$ )<sup>18</sup> and  $\text{SiO}_2$  ( $\sim 4 \times 10^{-21} \text{ cm}^2$ ). Evidence for such an unusually high cross-section has recently also been found by Shin<sup>6</sup> and Dal Negro<sup>19</sup>. The large cross-section may be due to the strong asymmetry of the local dielectric environment of the

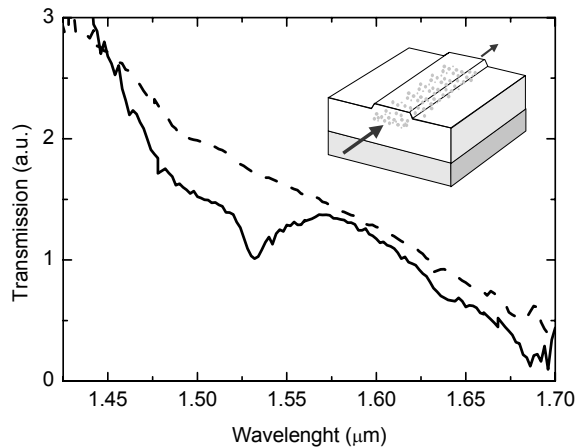


Figure 10. Transmitted signal in a 0.8 cm long Si nanocrystal based waveguides containing no Er (dotted line) and 1.3 at.% Er (solid line).

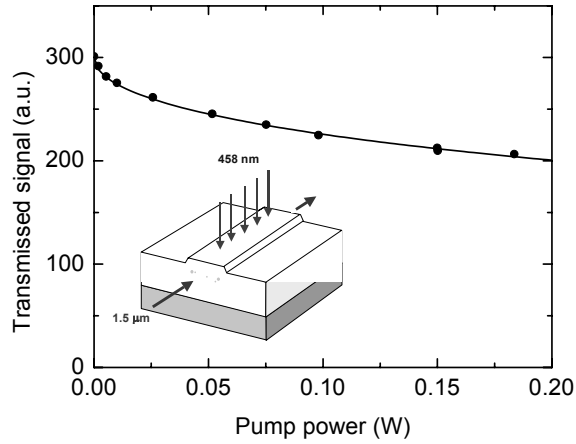


Figure 13. Transmitted signal at  $1.535 \mu\text{m}$  as a function of the applied continuous-wave  $458 \text{ nm}$  pump power as obtained from transmission measurements. The transmission at zero pump power includes the waveguide loss and Er ground state absorption. A sketch of the measurement configuration is included.

excitable  $\text{Er}^{3+}$  ions (note that in a perfectly symmetrical geometry the  ${}^4\text{I}_{13/2} \rightarrow {}^4\text{I}_{15/2}$  transition is parity-forbidden). This may also explain the relatively short luminescence lifetimes typically observed in this type of material. The high absorption cross-section also translates into a high emission cross-section,<sup>20,21</sup> which implies that high gain could be achieved in this material over a short length.

Figure 13 shows the transmitted signal at  $1.535 \mu\text{m}$  as a function of applied pump power at  $458 \text{ nm}$ . The same pump power dependence of the transmitted signal was observed at all wavelengths in the range  $1.4\text{-}1.7 \mu\text{m}$ . Apparently, instead of producing gain, optical pumping at  $458 \text{ nm}$  induces an absorption in the waveguide material. Figure 12

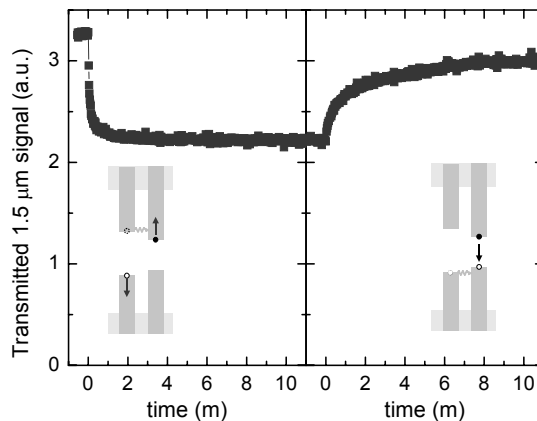


Figure 12. Time dependence of the  $1.5 \mu\text{m}$  transmission of an Er and Si nanocrystal doped optical waveguide excited with a  $458 \text{ nm}$  pump laser directly after pump switch-on (left panel) and pump switch-off (right panel).

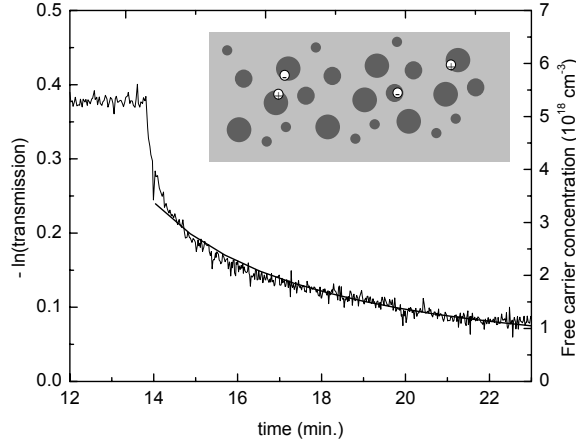


Figure 14. Time dependence of the calculated free-carrier concentration as discussed in section 4.6. The solid line is the solution of a rate equation model describing tunneling assisted bimolecular recombination of separated charges.

shows the time dependent transmission at 1.535  $\mu\text{m}$  after pump switch-on and switch-off. Note that after pump switch-off, the signal transmission returns to approximately its original value over a period of several *minutes*.

The observed transmission change could be due to free carrier absorption, a well-known effect in bulk silicon, in which free carriers absorb sub-bandgap radiation due to intra-band transitions.<sup>22</sup> The free carriers causing the absorption feature are not present in the form of quantum confined electron-hole pairs, since these recombine at a rate  $\sim 10^5 \text{ s}^{-1}$  at room temperature.<sup>5,23</sup> However, if the charge carriers are physically separated, their lifetime can increase by orders of magnitude.<sup>24</sup> Such separation could occur when during pumping one of the carriers in an electron-hole pair is ejected into a deep trap state near the nanocrystal surface, or when one of the carriers tunnels into a neighboring nanocrystal. The latter process is indicated schematically in Fig. 12.

The data in Fig. 13 can be used to estimate the pump-induced free carrier concentration. The maximum transmission reduction is approximately 1.7 dB, which corresponds to  $\alpha = -0.5 \text{ cm}^{-1}$  given the waveguide length of 8 mm. Taking  $\alpha(\text{cm}^{-1}) = \sigma_{\text{fc}} \times N_{\text{carr}} \times \Gamma$  with  $\sigma_{\text{fc}}$  a typical free carrier absorption cross-section in bulk Si at 1.5  $\mu\text{m}$  of  $7 \times 10^{-18} \text{ cm}^2$ ,  $N_{\text{carr}}$  the concentration of pump-induced free carriers in the waveguide, and  $\Gamma = 0.01$  the previously determined mode overlap with the Si nanocrystal doped region, we find  $N_{\text{carr}} = 0.7 \times 10^{19} \text{ cm}^{-3}$ , close to one carrier per nanocrystal. Note that this kind of pump induced absorption has not been observed in nanocrystal doped waveguides with lower excess Si concentrations,<sup>6</sup> possibly due to larger spacing between nanocrystals.

If the observed transmission reduction is indeed caused by single charges in separate nanocrystals, the recombination rate of these charges can be expected to decrease rapidly as the concentration of free charges decreases. This type of ‘bimolecular recombination’ can be described by the following rate equation:



$$\frac{dN_{carr}}{dt} = R - CN_{carr}^2 \quad (1)$$

with  $N_{carr}$  the concentration of *isolated* carriers,  $R$  the generation rate of isolated carriers, and  $C(\text{cm}^3\text{s}^{-1})$  a proportionality constant relating the recombination rate to the carrier concentration. If we assume a linear relation between the carrier generation rate and the pump power, the steady state solution gives  $N_{carr} \propto P^{1/2}$ . The solid line in Fig. 13 corresponds a transmission  $T = \exp(-\alpha(P)L)$  with  $L = 0.8$  cm the waveguide length and  $\alpha(\text{cm}^{-1}) = \sigma_{fc} \times N_{carr}(P) \times \Gamma$ , where we have taken  $N_{carr} \propto P^{1/2}$ . It is clear that the simple rate equation model can produce a good fit of the observed behavior.

Rate equation (1) can also be used to describe the time dependence of the pump induced absorption. In Fig. 14 we have plotted  $-\ln(T)$  over time after pump switch-off. Using the measured mode overlap, the bulk Si free-carrier absorption cross-section, and the known waveguide length, each value of  $-\ln(T)$  can be converted to  $N_{carr}$ , as indicated on the right axis. We should be able to describe this curve with equation (1) setting the pump rate to zero. In this pump-off condition the solution is of the form  $N_{carr}(t) = A/t$ , with  $A$  a scaling factor indicating the initial carrier concentration and the proportionality constant relating the recombination rate to the carrier concentration. This dependency is indicated in Fig. 14 by the drawn line. Except for an initial fast drop in the concentration, the curve describes the time dependent carrier concentration quite well. Despite the shortcomings of the model (no nanocrystal size dependence, no absorption by quantum confined *excitons*) the proposed process of tunneling-limited carrier recombination can indeed describe the main features of the observed pump-induced absorption.

## 5. The thin line between gain and loss

In this Section we discuss a simple gain model that includes the main processes determining the gain of an Er doped Si nanocrystal sensitized waveguide amplifier. The model includes stimulated emission by excited Er, absorption by Er in the ground state, absorption by free carriers present in nanocrystals not coupled to Er, and absorption by carriers present in Si nanocrystals that *are* coupled to Er. We have made the following assumptions: 1) the Er ions, the Si nanocrystals coupled to Er, and the Si nanocrystals not coupled to Er behave as three-level systems, 2) the chosen pump rate ( $R = 1000 \text{ s}^{-1}$ ) is sufficient to keep excitable Er ions inverted, 3) each nanocrystal can keep at most 1 Er ion in the excited state, 4) excitons in Si nanocrystals not coupled to Er have a decay time of 10  $\mu\text{s}$ , and excitons in Si nanocrystals that are coupled to Er have a 1  $\mu\text{s}$  decay time, 5) the cross-section for free carrier absorption in Si nanocrystals is a factor 10 lower ( $\sigma_{fc} = 10^{-18} \text{ cm}^2$ ) than in bulk Si due to a the reduced density of states in the conduction band and the valence band, 6) charge cannot tunnel to neighboring nanocrystals, 7) all nanocrystals contain 1000 atoms (diameter  $\sim 3$  nm), 8) the sample contains  $10^{19}$  nanocrystals/ $\text{cm}^3$ , and 9) the optical mode overlap is 100%. Although one can argue about the accuracy of some of these assumptions, the model does allow us to illustrate the main processes that influence the gain.

The calculations were done for ion implanted samples in which the Er ions are distributed statistically throughout the material. For example, at a concentration of 0.1 at.%, on average each 1000-atom nanocrystal will contain 1 Er ion, but due to the statistical Er

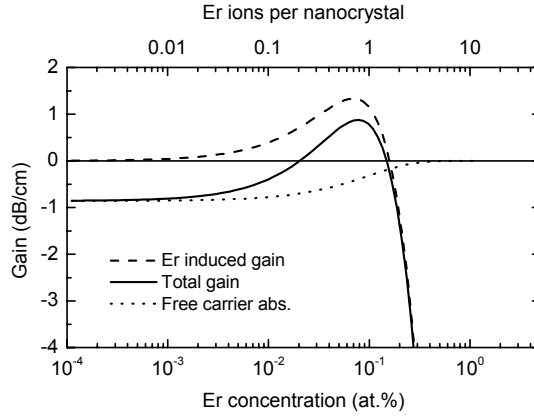


Figure 15. Gain calculations on an Er doped Si nanocrystal sensitized waveguide amplifier, showing the contributions of free-carrier absorption and Er induced stimulated emission and ground state absorption.

distribution some nanocrystals will contain zero Er ions, and some will contain more than one Er ion. Figure 15 shows the individual contributions to the optical gain at 1.54  $\mu\text{m}$  as a function of Er concentration. The dashed line shows the contribution of the Er ions. At low concentration, no Er is available, resulting in a gain of 0 dB. As the Er concentration is increased, the gain increases correspondingly. As the concentration approaches 1 Er ion per nanocrystal, in some nanocrystals the maximum excitable Er concentration is exceeded. These excess Er ions cannot be inverted, resulting in ground state absorption (i.e. negative gain) occurring at 1.54  $\mu\text{m}$ . This effect becomes more important at increasing Er concentration.

The dotted line shows the contribution of absorption by excitons in nanocrystals that are not coupled to Er. At low concentration, all nanocrystals have a decay time of 10  $\mu\text{s}$ . At the pump rate of 1000  $\text{s}^{-1}$  this produces an exciton in 1% of the nanocrystals, or  $10^{17} \text{ cm}^{-3}$ . Consequently, the free carrier absorption coefficient is  $\alpha_{\text{fc}}(\text{cm}^{-1})=2\times\sigma_{\text{fc}}\times 10^{17} \text{ cm}^{-3}=0.2 \text{ cm}^{-1}$  or 0.87 dB/cm, where the factor 2 is due to the fact that an exciton consists of 2 free charges. As the Er concentration is increased, the chance for a nanocrystal to be coupled to Er increases, which reduces the exciton lifetime to 1  $\mu\text{s}$ , resulting in reduced free carrier absorption. Since the lifetime of coupled nanocrystals is a factor ten less than that of non-coupled nanocrystals, the maximum absorption due to excitons generated in coupled nanocrystals at this pump rate is approximately  $0.1\times 0.87 \text{ dB/cm} \approx 0.1 \text{ dB/cm}$ . This contribution is negligible in Fig. 15 but could become significant at higher pump powers. The solid line shows the sum of the two gain contributions, illustrating that net gain can be achieved only in a narrow Er concentration range.

The gain calculations in Fig. 15 can be considered as a worst case scenario in several ways. For one thing, there is as yet no clear consensus on the maximum number of excitable Er ions per nanocrystal, the estimates ranging from 1 to 100. A higher concentration of excitable Er will dramatically increase the maximum attainable gain. Additionally, the free-carrier absorption observed at low Er concentration is caused by nanocrystals that are not coupled to Er. It seems feasible that a fabrication process can be developed that prevents the formation of uncoupled nanocrystals. One final point of

concern may be the signal power. The gain calculations were done in a relatively low pump power regime. When high intensity signals are amplified, higher pump rates will be necessary. It seems likely that given the finite lifetime of excitons in nanocrystals coupled to Er, a significant free-carrier concentration may be built up at high pump power. This could put an upper limit on the signal strength that can be amplified.

## 6. Conclusions

We have shown that Er excitation via Si nanocrystals can satisfy all requirements to be used in broad-band pumped integrated optical waveguide amplifiers and lasers operating at 1.5  $\mu\text{m}$ . In particular, it was shown that (a) the Si nanocrystal absorption cross-section at visible wavelengths is several orders of magnitude larger than direct Er optical absorption cross-sections, allowing for low pump-thresholds and unconventional pumping geometries, (b) the energy transfer quantum efficiency is larger than 55%, (c) energy transfer to Er occurs under visible excitation in a broad wavelength band, allowing for broad-band pumping using cost-effective light sources such as LEDs, (d) the energy transfer rate is larger than  $10^6 \text{ s}^{-1}$ , in principle allowing for the generation of several hundreds of milliwatts 1.5  $\mu\text{m}$  power in a 1 cm long waveguide, (e) concentrations of optically active Er in excess of  $10^{19} \text{ cm}^{-3}$  may be excited via nanocrystals, (f) optical pumping of nanocrystal sensitized can introduce a broad absorption in the infrared, possibly due to free-carrier absorption, and (g) the direct optical absorption cross-section of Er in a nanocrystal doped medium is enhanced by a factor  $\sim 10$  compared to the absorption cross-section in typical  $\text{SiO}_2$  based hosts. Gain calculations taking into account all these observations suggest that net gain at 1.5  $\mu\text{m}$  in excess of 1 dB/cm can be achieved in Er doped Si nanocrystal sensitized optical waveguide amplifiers.

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